# Crystal Structure of $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathrm{P}_{4} \mathrm{O}_{42}$ : A Fluorite-Related Superstructure 

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Received October 20, 1999; in revised form January 11, 2000; accepted January 20, 2000

Single crystals of a new oxyphosphate of bismuth and lead, $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathbf{P}_{4} \mathrm{O}_{42}$, were studied using short-wavelength synchrotron X-ray radiation. The compound is monoclinic, $I 2 / m, a=$ $11.885(2) \AA, b=11.540(2) \AA, c=15.636(3) \AA, \quad \beta=90.23(3)^{\circ}$, $Z=2$. The crystal is twinned by pseudo-merohedry. The structure refinement converged to $R=0.0366$. There are five crystallographically independent Bi atoms. The oxygen coordination polyhedron around $\mathrm{Bi}(1)$ is a slightly distorted cube with $\mathrm{Bi}-\mathrm{O}$ bonds varying from $2.381(11)$ to $2.662(10) \AA$. The other four Bi atoms are in a severely distorted oxygen environment with bond lengths varying from $2.102(5)$ to $\mathbf{3 . 1 6 ( 2 )} \AA$. A greater variation of the oxygen coordination polyhedra exists around the three crystallographically independent $\mathbf{P b}$ atoms changing from a somewhat elongated cube to a monocapped pentagonal bipyramid; bond distances vary from $2.13(2)$ to $3.38(2) \AA$. Of 20 crystallographically independent oxygen ions 11 are disordered. The structure consists of two $\mathrm{Bi}-\mathrm{O}$ layers parallel to (101). They are interconnected perpendicular to [010] by layers consisting of $\mathbf{P b}$ and $\mathrm{PO}_{4}$. The two crystallographically independent $\mathrm{PO}_{4}$ moieties are disordered. This compound displays a $\delta-\mathrm{Bi}_{2} \mathrm{O}_{3}$ fluorite superstructure. The locations of the nonbonded lone-pair electrons were calculated and vary from 0 to $1.1 \AA$. These displacements from the cation nucleus are indicative of the extent of $s-p$ hybridization, where zero displacement represents the spherical nonbonded $\mathbf{6 s} s^{\mathbf{2}}$ orbital. The extent of hybridization is a function of the distortion of the oxygen coordination polyhedron around the cation. © 2000 Academic Press

Key Words: $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathbf{P}_{4} \mathrm{O}_{42} ;$ structure; disorder; $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathbf{P}_{4} \mathrm{O}_{42}$; lone-pair electrons and coordination polyhedra, influence of.

## INTRODUCTION

$\mathrm{Bi}_{2} \mathrm{O}_{3}$ exhibits four different structure types (1-3). The high-temperature fluorite type $\delta-\mathrm{Bi}_{2} \mathrm{O}_{3}$ is well known as one of the best anionic conductors due to oxygen vacancies in the structure (4). The discovery that $\mathrm{Bi}_{4} \mathrm{~V}_{2} \mathrm{O}_{11}$ was a good oxide ion conductor (5) spawned intensive research on this and related phases (6). An extensive review by Boivin and Mairesse discusses recent developments in fast oxide ion
conductors (7). Many studies of bismuth-based oxides have been carried out and numerous compounds with high bismuth content have been reported as having a fluorite superstructure (8-13). The pseudo-ternary systems $\mathrm{PbO}-\mathrm{Bi}_{2} \mathrm{O}_{3}-$ $M_{2} \mathrm{O}_{5}(M=\mathrm{P}, \mathrm{V}, \mathrm{As})$ have been extensively investigated (14-20) and recently we discovered new compounds with the formula $\mathrm{Pb}_{5} \mathrm{Bi}_{18} M_{4} \mathrm{O}_{42}(M=\mathrm{P}, \mathrm{V}$, As). They are related to a fluorite type superstructure. This paper reports the structure of the new phase $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathrm{P}_{4} \mathrm{O}_{42}$.

## EXPERIMENTAL

$\mathrm{Bi}_{2} \mathrm{O}_{3}$ (Strem 99.9\%) and PbO (Alfa 99.97\%) were preheated overnight at $600^{\circ} \mathrm{C}$ to remove any carbonate. They were mixed with $\left(\mathrm{NH}_{4}\right)_{2} \mathrm{HPO}_{4}($ Strem $99+\%)$ in stoichiometric ratios to obtain phases $\mathrm{Pb}_{4+x} \mathrm{Bi}_{19-x} \mathrm{P}_{4} \mathrm{O}_{y}(x=0,1)$. These stoichiometric compositions were chosen because the exploration of the tie line $\mathrm{PbBiPO}_{5}-\mathrm{Bi}_{2} \mathrm{O}_{3}$ (20) led to the discovery of a new phase near those stoichiometries. The mixtures containing the phosphate were slowly heated at $300^{\circ} \mathrm{C}$ to decompose $\left(\mathrm{NH}_{4}\right)_{2} \mathrm{HPO}_{4}$.

Initially the samples were examined by X-ray diffraction with a Guinier-de Wolf camera and $\mathrm{CuK} \alpha$ radiation. Subsequently, powder patterns were recorded with $\mathrm{CuK} \alpha$ radiation using an automated D5000 Siemens diffractometer equipped with a diffracted beam graphite monochromator. Differential thermal analysis (DTA) was performed on a Linseis L62 instrument. Density measurements were done with an automated Micrometritics Accupyc 1330 gas picnometer equipped with a $1-\mathrm{cm}^{3}$ cell. The possibility of phase transitions was checked with a high-temperature X-ray diffraction (HTXRD) Guinier-Lenné camera at a heating rate of $20^{\circ} \mathrm{C} \mathrm{min}^{-1}$.

The melting point of $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathrm{P}_{4} \mathrm{O}_{42}$ is $930^{\circ} \mathrm{C}$ and single crystals were obtained by melting the powder in a gold crucible. The powder was heated in air to $970^{\circ} \mathrm{C}$ for 3 h , slowly cooled at a rate of $1.2^{\circ} \mathrm{C} / \mathrm{h}$ to $850^{\circ} \mathrm{C}$, and then furnace cooled to room temperature. The crystals are light yellow needles. Single crystals were selected under a polarizing microscope and Weissenberg photographs were obtained.

The reciprocal lattice photographs indicated an apparent I-centered orthorhombic cell. The data for the structure determination were recorded with synchrotron radiation, $\lambda=0.1602 \AA$, as part of a DuPont program to study the general effect of absorption on crystal structure determinations. The data were collected using an MAR-CCD detector on beamline 5BMD at the Advanced Photon Source, Argonne National Laboratory. A total of 64 frames were collected: 300 sec per frame, 3 deg oscillation, $\lambda=0.1602 \AA$, at a crystal-to-detector distance of 200 mm . The program DENZO (21) was used to analyze the frames yielding a Ccentered monoclinic unit cell, $a=19.602 \AA, b=11.540 \AA$, $c=11.885 \AA, \beta=127.09^{\circ}$. The intensity data were then extracted and merged, $R(F)=0.046$ and $R\left(F^{2}\right)=0.039$. No absorption correction was made since $\mu_{1}=1.67 \mathrm{~mm}^{-1}$. The unit cell was reoriented to yield the nonstandard monoclinic $I$-centered unit cell, $a=11.885(2) \AA, \quad b=11.540(2) \AA$, $c=15.636(3) \AA, \beta=90.23(3)^{\circ}$, to emphasize the nearly orthogonal $\beta$ angle. The crystal data and refinement are

TABLE 1
Crystal Data and Refinement for $\mathbf{P b}_{5} \mathbf{B i}_{18} \mathbf{P}_{\mathbf{4}} \mathbf{O}_{\mathbf{4 2}}$

| Crystallographic Data |  |  |
| :---: | :---: | :---: |
| Formula | $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathrm{P}_{4} \mathrm{O}_{42}$ |  |
| Color | Light yellow |  |
| Crystal system | Monoclinic |  |
| Space group | I2/m |  |
| $a($ A $)$ | 11.885(2) |  |
| $b(\AA)$ | 11.540(2) |  |
| $c(\AA)$ | 15.636(3) |  |
| $\beta\left({ }^{\circ}\right)$ | 90.23(3) |  |
| Volume ( $\AA^{3}$ ) | 2144.5(7) |  |
| Z | 2 |  |
| Formula weight (g/mol) | 5593.47 |  |
| Measured/calculated density ( $\mathrm{g} / \mathrm{cm}^{3}$ ) | 8.62/8.68 |  |
| Intensity Collection |  |  |
| Radiation ( A ) | 0.1602 |  |
| $\mu_{1}\left(\mathrm{~mm}^{-1}\right)$ | 1.67 |  |
| Temperature ( ${ }^{\circ} \mathrm{C}$ ) | 298 |  |
| $\theta$ range | 0.93-8.08 ${ }^{\circ}$ |  |
| Data collected | $\begin{aligned} & -16 \leq h \leq 20, \\ & -15 \leq l \leq 27 \end{aligned}$ | $0 \leq k \leq 20$, |
| No. of reflections measured | 6258 |  |
| No. of uniqaue reflections $I>2 \sigma(I)$ | 3498 |  |
| $R\left(F^{2}\right)_{\text {int }}$ | 0.039 |  |
| Refinement |  |  |
| Parameters varied | 142 |  |
| Refinement method | Least squares on $F^{2}$ |  |
| Twin function | 100, $0-10,00-1$ |  |
| Twin fraction | 0.26 |  |
| $R(F)$ obs/all | 0.0366/0.0744 |  |
| $R_{\text {w }}\left(F^{2}\right)$ obs/all | 0.0787/0.1568 |  |
| $w=1 /\left(\sigma^{2}\left(F_{\mathrm{o}}^{2}\right)+(X P)^{2}+0.00 P\right) \text { with }$ |  |  |
| GOF obs/all | 0.682/0.559 |  |
| Max., min, $\Delta \rho$, e $\AA^{-3}$ | $7.868,-7.971$ |  |

TABLE 2
Powder X-Ray Diffraction Diagram of $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathbf{P}_{\mathbf{4}} \mathbf{O}_{\mathbf{4 2}}$

| $h$ | $k$ | $l$ | $d_{\text {obs }}(\AA)$ | $d_{\text {cal }}(\AA)$ | $I(\%)$ | $h$ | $k$ | $l$ | $d_{\text {obs }}(\AA)$ | $d_{\text {cal }}(\AA)$ | $I(\%)$ |
| ---: | ---: | ---: | ---: | ---: | ---: | ---: | ---: | ---: | ---: | ---: | ---: |
| 1 | 0 | 1 | 9.491 | 9.489 | 8 | 1 | 4 | -3 | 2.482 | 2.482 | $<1$ |
| 0 | 1 | 1 | 9.324 | 9.330 | 1 | 2 | 4 | 2 | 2.473 | 2.473 | $<1$ |
| 1 | 1 | 0 | 8.296 | 8.320 | 1 | 2 | 0 | -6 | 2.403 | 2.404 | $<1$ |
| 0 | 0 | 2 | 7.872 | 7.859 | $<1$ | 0 | 2 | 6 | 2.387 | 2.387 | $<1$ |
| 0 | 2 | 0 | 5.787 | 5.797 | 3 | 4 | 3 | -1 | 2.339 | 2.338 | 1 |
| 1 | 1 | -2 | 5.738 | 5.723 | 2 | 4 | 3 | 1 | 2.332 | 2.333 | $<1$ |
| 1 | 1 | 2 | 5.708 | 5.704 | 1 | 0 | 4 | 4 | 2.317 | 2.317 | $<1$ |
| 2 | 1 | 1 | 5.028 | 5.024 | 7 | 3 | 2 | -5 | 2.276 | 2.275 | $<1$ |
| 1 | 2 | 1 | 4.945 | 4.947 | $<1$ | 3 | 3 | -4 | 2.271 | 2.270 | $<1$ |
| 1 | 0 | 3 | 4.794 | 4.790 | 1 | 2 | 3 | -5 | 2.261 | 2.261 | 1 |
| 0 | 1 | 3 | 4.771 | 4.775 | 6 | 2 | 3 | 5 | 2.255 | 2.255 | 2 |
| 3 | 0 | -1 | 3.863 | 3.865 | $<1$ | 5 | 1 | -2 | 2.244 | 2.246 | $<1$ |
| 3 | 1 | 0 | 3.768 | 3.766 | 3 | 2 | 2 | 6 | 2.212 | 2.213 | $<1$ |
| 2 | 1 | 3 | 3.721 | 3.721 | $<1$ | 0 | 1 | 7 | 2.204 | 2.205 | $<1$ |
| 1 | 2 | -3 | 3.702 | 3.700 | $<1$ | 4 | 2 | 4 | 2.195 | 2.196 | $<1$ |
| 1 | 1 | -4 | 3.556 | 3.558 | 2 | 5 | 2 | -1 | 2.190 | 2.189 | $<1$ |
| 1 | 1 | 4 | 3.550 | 3.549 | 1 | 5 | 2 | 1 | 2.186 | 2.186 | $<1$ |
| 3 | 1 | -2 | 3.408 | 3.403 | $<1$ | 2 | 4 | -4 | 2.174 | 2.175 | 1 |
| 1 | 3 | -2 | 3.333 | 3.333 | 4 | 2 | 4 | 4 | 2.171 | 2.171 | $<1$ |
| 1 | 3 | 2 | 3.330 | 3.329 | 1 | 3 | 1 | -6 | 2.156 | 2.155 | 1 |
| 2 | 0 | -4 | 3.289 | 3.291 | 1 | 3 | 1 | 6 | 2.146 | 2.146 | $<1$ |
| 3 | 2 | -1 | 3.216 | 3.216 | 1 | 2 | 5 | -1 | 2.142 | 2.142 | 1 |
| 3 | 0 | -3 | 3.178 | 3.178 | 76 | 3 | 4 | 3 | 2.137 | 2.137 | 1 |
| 3 | 0 | 3 | 3.162 | 3.163 | 100 | 0 | 5 | 3 | 2.120 | 2.120 | $<1$ |
| 0 | 3 | 3 | 3.109 | 3.110 | 94 | 2 | 1 | -7 | 2.072 | 2.072 | $<1$ |
| 1 | 0 | -5 | 3.042 | 3.044 | 4 | 2 | 1 | 7 | 2.067 | 2.065 | $<1$ |
| 1 | 0 | 5 | 3.036 | 3.037 | 2 | 1 | 2 | -7 | 2.064 | 2.064 | 2 |
| 0 | 4 | 0 | 2.898 | 2.898 | 1 | 5 | 1 | -4 | 2.016 | 2.015 | $<1$ |
| 2 | 2 | -4 | 2.863 | 2.862 | $<1$ | 5 | 1 | 4 | 2.006 | 2.006 | 2 |
| 2 | 2 | 4 | 2.849 | 2.852 | 1 | 6 | 0 | 0 | 1.991 | 1.991 | 16 |
| 4 | 1 | 1 | 2.843 | 2.842 | 1 | 0 | 4 | 6 | 1.944 | 1.944 | 1 |
| 4 | 0 | -2 | 2.796 | 2.797 | 8 | 0 | 6 | 0 | 1.933 | 1.932 | 15 |
| 4 | 0 | 2 | 2.788 | 2.787 | 8 | 3 | 3 | -6 | 1.908 | 1.908 | 17 |
| 3 | 3 | 0 | 2.773 | 2.773 | 45 | 3 | 3 | 6 | 1.901 | 1.901 | 19 |
| 0 | 0 | 6 | 2.619 | 2.620 | 22 | 4 | 3 | -5 | 1.893 | 1.893 | 3 |
| 4 | 1 | -3 | 2.536 | 2.537 | $<1$ | 4 | 3 | 5 | 1.886 | 1.886 | 2 |
| 4 | 1 | 3 | 2.528 | 2.527 | $<1$ |  |  |  |  |  |  |
|  |  |  |  |  |  |  |  |  |  |  |  |

Note. Lattice parameters: $a=11.946(3) \AA, \quad b=11.592(3) \AA, \quad c=$ 15.719(3) $\AA, \beta=90.29(1)^{\circ} ; \mathrm{Cu} K \alpha$.
shown in Table 1. The structure was solved using the direct method and refined by full-matrix least squares $(22,23) . \mathrm{Pb}$, Bi , and P atoms were refined anisotropically, while O atoms were refined isotropically. The least-squares refinement was based on 3498 unique reflections $I>2 \sigma(I)$ and converged to $R=0.0366$.

## RESULTS

All observed lines of the powder pattern of $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathrm{P}_{4} \mathrm{O}_{42}$ were indexed. The unit cell parameters were refined by least squares. The indexed powder pattern and lattice parameters are shown in Table 2. There is a significant difference between the single-crystal and powder diffraction cell parameters. The synchrotron wavelength was determined using
the $\mathrm{Au} K$ absorption edge and is accurate. However, the crystal-to-detector distance is less precise and a small error in that distance will greatly impact the cell parameters. We attempted to quantify the $\mathrm{Pb} / \mathrm{Bi}$ ratio using standardless X-ray energy dispersive techniques (EDX) but the results were not definitive. $\mathrm{The} \mathrm{Pb} / \mathrm{Bi}$ ratios varied from 0.25 to 0.29 . The synthesis of a stoichiometric mixture to yield $\mathrm{Pb}_{6} \mathrm{Bi}_{17} \mathrm{P}_{4} \mathrm{O}_{y}$ seemed to have the same powder X -ray diffraction pattern as the $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathrm{P}_{4} \mathrm{O}_{42}$ phase but $\mathrm{Pb}_{4} \mathrm{Bi}_{19} \mathrm{P}_{4} \mathrm{O}_{7}$ showed weak lines of a second phase. We believe that $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathrm{P}_{4} \mathrm{O}_{42}$ exhibits a slight range of solid solubility and that may also account for the differences in the lattice parameters obtained from the single crystal and the powder.

The measured density of the phosphate compound, $\rho=$ $8.62 \mathrm{~g} / \mathrm{cm}^{3}$, yields two formula weights of $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathrm{P}_{4} \mathrm{O}_{42}$ per unit cell. No indications of a phase transition were observed on DTA and HTXRD. When the data were merged on the basis of an orthorhombic symmetry $R_{\mathrm{int}}$ was unacceptably high while $R\left(F^{2}\right)_{\text {int }}$ is 0.039 for a monoclinic system. The space group $I 2$ was chosen initially and 13 heavy atoms resulting from the direct method and subsequent Fourier maps were introduced, yielding $R=0.165$. Including parameters of two P atoms obtained from a difference electron density map dropped $R$ to 0.14 . A center of symmetry was now identified and the refinement was continued in the centrosymmetric group $I 2 / m$. At this stage $R$ was 0.15 . The oxygen atomic positions were obtained from successive Fourier difference maps. Fourteen oxygen atoms were found. The displacement parameters were refined anisotropically for $\mathrm{Pb}, \mathrm{Bi}$, and P , and isotropically for oxygen atoms. The refinement converged to $R=0.064$ but some of the oxygen atoms required for the $\mathrm{PO}_{4}$ groups were still missing.

Monoclinic cells with $\beta \cong 90^{\circ} \mathrm{C}$ often twin by a rotation of $180^{\circ}$ about the $a$ axis, ( $100,0-10,00-1$ ), displaying
a pseudo-orthorhombic cell. This twinning function was introduced in the refinement, dropping the $R$ value to 0.0377 . The fraction of the twin is $26 \%$. At this point some oxygen atoms were still missing from the $\mathrm{PO}_{4}$ groups. The examination of difference Fourier maps revealed some additional peaks of low density at reasonable positions from phosphorus to form tetrahedra and they were assigned to oxygen atoms. It became evident that multiple tetrahedra were centered on the two crystallographically independent phosphorus atoms. The very low electron density peaks are due to the partial site occupancies (sof) by oxygen. $\mathrm{P}(1)$ is bonded to $\mathrm{O}(9), \mathrm{O}(10), 2 \mathrm{O}(11), 2 \mathrm{O}(16)$, and $\mathrm{O}(17)$. The sof for $\mathrm{O}(9)$ is 1 . The sum of the sof's of the other oxygen atoms was restrained to form two $\mathrm{PO}_{4}$ tetrahedra, Fig. 1. $\mathrm{P}(2)$ is bonded to $\mathrm{O}(12), 2 \mathrm{O}(13), \mathrm{O}(14), \mathrm{O}(15), 2 \mathrm{O}(18), 2 \mathrm{O}(19)$, and $\mathrm{O}(20)$. The coordinates of $\mathrm{O}(14)$ and $\mathrm{O}(20)$ were fixed because they were ill-behaved during least-squares refinements. The bond distance $\mathrm{P}(2)-\mathrm{O}(15)$ was restrained to a value greater than $1.4 \AA$. Without such a restraint leastsquares refinements yielded values of about $1.35 \AA$. Restraints were also applied to the sof's of oxygen atoms around $\mathrm{P}(2)$ so that the total number of oxygen atoms yielded three $\mathrm{PO}_{4}$ moieties, Fig. 1. Isotropic displacement parameters of oxygen were obtained by least-squares refinements. The last cycle of refinement yielded $R=0.0366$. Table 3 lists the final atomic parameters, Table 4 the anisotropic displacement parameters, and Table 5 the interatomic bond lengths and angles, the bond valences, and their sums (24).

## LONE-PAIR LOCALIZATION

The interplay among lone-pair electrons, stereochemistry, and crystal structure has long been of interest to crystallographers and solid-state chemists (25-27). Verbaere et al. (28) developed the theory enabling the calculation for


FIG. 1. Oxygen environment around $\mathrm{P}(1)$ and $\mathrm{P}(2)$. All oxygen atoms are shown to illustrate the disorder. The dashed and full bonds show the two tetrahedra around $\mathrm{P}(1)$. $\mathrm{O}(9)$ is above $\mathrm{P}(1)$ and $\mathrm{O}(15)$ above $\mathrm{P}(2)$.

TABLE 3
Atomic Coordinates and Equivalent Displacement Parameters, $\AA^{2}$, for $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathbf{P}_{4} \mathbf{O}_{42}$

| Atom | Occupancy | $x$ | $y$ | $z$ | $U_{\text {eq }}{ }^{a}$ |
| :---: | :---: | :---: | :---: | :---: | :---: |
| Bi(1) | 1 | 0.5 | 0.17676(4) | 0.5 | 0.0380(2) |
| Bi(2) | 1 | $0.16189(4)$ | 0.16729(3) | 0.17434(3) | 0.0219(1) |
| Bi(3) | 1 | $0.65596(4)$ | 0.33241(3) | 0.32187(3) | 0.0208(1) |
| Bi(4) | 1 | 0.68436(3) | 0.33988(2 | -0.00402(4) | 0.0198(1) |
| Bi(5) | 1 | -0.00413(5) | 0.33894(3) | 0.34941(2) | 0.0191(1) |
| $\mathrm{Pb}(1)$ | 1 | 0 | 0 | 0 | 0.0323(2) |
| $\mathrm{Pb}(2)$ | 1 | 0.66636(8) | 0 | 0.33797(6) | 0.0380(2) |
| $\mathrm{Pb}(3)$ | 1 | 0.34985(8) | 0 | 0.32457(6) | 0.0373(2) |
| $\mathrm{P}(1)$ | 1 | 0.4984(5) | 0.5 | -0.1572(2) | 0.0163(6) |
| $\mathrm{P}(2)$ | 1 | -0.2981(3) | 0 | 0.0026(4) | 0.0190(7) |
| $\mathrm{O}(1)$ | 1 | 0.765(1) | 0.5 | 0.015(1) | 0.020(2) |
| $\mathrm{O}(2)$ | 1 | 0.503(2) | 0 | 0.396(1) | 0.036(2) |
| $\mathrm{O}(3)$ | 1 | 0.183(1) | 0.318(1) | 0.097(1) | 0.026(2) |
| $\mathrm{O}(4)$ | 1 | 0.811(1) | 0.304(1) | 0.090(1) | 0.027(2) |
| $\mathrm{O}(5)$ | 1 | 0.012(2) | 0.5 | 0.287(1) | 0.026(2) |
| O(6) | 1 | 0.003(2) | 0.169(1) | 0.101(1) | 0.069(3) |
| $\mathrm{O}(7)$ | 1 | 0.635(1) | 0.187(1) | 0.240(1) | 0.041(2) |
| $\mathrm{O}(8)$ | 1 | 0.133(2) | 0.295(1) | 0.270(1) | 0.064(3) |
| $\mathrm{O}(9)$ | 1 | 0.608(2) | 0.5 | -0.110(2) | 0.073(6) |
| $\mathrm{O}(10)$ | 0.77(2) | 0.518(3) | 0.5 | 0.752(2) | 0.067(6) |
| O11) | 0.77(2) | 0.430(2) | 0.605(1) | 0.864(1) | 0.051(3) |
| $\mathrm{O}(12)$ | 0.84(2) | 0.635(2) | 0 | 0.086(2) | 0.046(4) |
| $\mathrm{O}(13)$ | 0.54(2) | 0.679(2) | 0.891(2) | 0.951(2) | $0.027(3)$ |
| $\mathrm{O}(14)$ | 0.54(2) | 0.821 | 0 | 0.031 | 0.103(15) |
| $\mathrm{O}(15)$ | 0.46(2) | 0.616(5) | 0 | 0.939(4) | 0.083(14) |
| $\mathrm{O}(16)$ | 0.24(2) | 0.995(8) | $0.102(5)$ | 0.718(5) | 0.089(17) |
| $\mathrm{O}(17)$ | 0.24(2) | 0.606(5) | 0.5 | 0.098(4) | $0.021(8)$ |
| $\mathrm{O}(18)$ | 0.29(2) | 0.261(4) | 0.393(3) | 0.484(4) | 0.065(11) |
| $\mathrm{O}(19)$ | 0.16(2) | 0.177(7) | 0.397(5) | 0.552(5) | 0.044(15) |
| $\mathrm{O}(20)$ | 0.16(2) | 0.809 | 0 | 0.952 | 0.081(38) |

${ }^{a} U_{\text {eq }}$ is defined as $\frac{1}{3}$ of the trace of the diagonalized $U_{i j}$ tensor.
localizing the lone-pair electrons in a crystal structure. It is based on the local electric field calculation in the whole crystal using Ewald's method (29) and was recently success-

TABLE 4
Anisotrophic Displacement Parameters for $\mathbf{B i}, \mathbf{P b}$, and $\mathbf{P}, \AA^{2}$

| Atom | $U_{11}$ | $U_{22}$ | $U_{33}$ | $U_{23}$ | $U_{13}$ | $U_{12}$ |
| :--- | :---: | :---: | :---: | :---: | ---: | :--- |
| $\operatorname{Bi}(1)$ | $0.0481(5)$ | $0.0211(2)$ | $0.0447(4)$ | 0 | $-0.0061(5)$ | 0 |
| $\operatorname{Bi}(2)$ | $0.0252(2)$ | $0.0156(2)$ | $0.0248(2)$ | $0.0037(2)$ | $-0.0077(2)$ | $0.0003(2)$ |
| $\operatorname{Bi}(3)$ | $0.0214(2)$ | $0.0170(2)$ | $0.0238(2)$ | $0.0046(2)$ | $0.0035(2)$ | $0.0016(2)$ |
| $\operatorname{Bi}(4)$ | $0.0179(2)$ | $0.0173(1)$ | $0.0243(2)$ | $-0.0010(2)$ | $-0.0008(2)$ | $-0.0033(2)$ |
| $\operatorname{Bi}(5)$ | $0.0187(2)$ | $0.0218(1)$ | $0.0167(2)$ | $0.0044(2)$ | $-0.0008(2)$ | $-0.0005(2)$ |
| $\operatorname{Pb}(1)$ | $0.0605(7)$ | $0.0148(2)$ | $0.0217(3)$ | 0 | $0.0098(6)$ | 0 |
| $\operatorname{Pb}(2)$ | $0.0353(4)$ | $0.0380(3)$ | $0.0408(5)$ | 0 | $0.0150(4)$ | 0 |
| $\operatorname{Pb}(3)$ | $0.0323(4)$ | $0.0445(3)$ | $0.0349(4)$ | 0 | $-0.0158(4)$ | 0 |
| $\mathrm{P}(1)$ | $0.019(2)$ | $0.0134(8)$ | $0.017(2)$ | 0 | $-0.002(2)$ | 0 |
| $\mathrm{P}(2)$ | $0.016(2)$ | $0.0140(8)$ | $0.027(2)$ | 0 | $0.001(2)$ | 0 |

[^0]fully applied to several materials $(26,30,31)$. It was incorporated in the computer program HYBRIDE (31). The required mean ionicity for the three cations was calculated for $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathrm{P}_{4} \mathrm{O}_{42}$ from the electronegativity differences using the formula
$$
M-\mathrm{O}=1-\exp \left[-\left(X_{\mathrm{O}}-X_{\mathrm{M}}\right)^{2} / 4\right]
$$

These values provided the oxidation state +1.22 for $\mathrm{Pb}^{2+}$, 1.7 for $\mathrm{Bi}^{3+}$, and +2.0 for $\mathrm{P}^{5+}$. The oxygen atoms were given the balancing charge -1.06 to ensure electroneutrality within the lattice. The positions of the lone-pair electrons of Pb and Bi are shown in Table 6. These distances reflect the displacement of the electron density of the $s-p$ hybridized orbital from the nucleus. A zero distance means that the spherical $6 s^{2}$ orbital is centered on the nucleus as expected and no hybridization has occurred. With increasing displacements from the nucleus the $s-p$ hybridization increases and it is directly correlated with the distortion of the oxygen polyhedron around the cation.

## DISCUSSION

The Pb and Bi atoms display a number of different coordination polyhedra. $\mathrm{Bi}(1)$ is bonded to eight oxygen atoms that have full site occupancies at the distances shown in Table 5 . They form a slightly distorted cube around $\mathrm{Bi}(1)$. Fig. 2A. The calculated position of the electron lone pair, $0.21 \AA$, is very close to the nucleus. We ascribe this proximity to the nucleus as due to the electrostatic repulsion by the nearly regular array of oxygen anions around $\operatorname{Bi}(1)$. The valence bond sum of 2.32 may be indicative of a mixed $\mathrm{Bi} / \mathrm{Pb}$ site occupancy. $\mathrm{Bi}(2)$ is coordinated by the fully occupied sites of $\mathrm{O}(3), \mathrm{O}(6)$, and $2 \mathrm{O}(8)$, to one of the disordered $\mathrm{O}(10)$ sof 0.76 , or $\mathrm{O}(16)$ sof 0.24 , and to one of the disordered $\mathrm{O}(13)$ sof $0.54, \mathrm{O}(18)$ sof 0.29 , or $\mathrm{O}(20)$ sof 0.16 . The sof's were restrained in the least-squares refinements to add to unity for each set of disordered oxygen ions. The coordination polyhedron using only $\mathrm{O}(16)$ and $\mathrm{O}(18)$ is shown in Fig. 2A. Five anions form a nearly rectangular pyramid with $\mathrm{O}(3)$ occupying the apex. The lone-pair electrons push the sixth atom, $\mathrm{O}(16)$, away from what otherwise would be a position to complete an octahedron. The lone pair is $0.41 \AA$ from the nucleus. The use of other combinations of partially occupied oxygen sites creates similar polyhedra. This is also the case for the other Bi coordination polyhedra. The coordination polyhedron around $\mathrm{Bi}(3)$ is very similar to that of $\operatorname{Bi}(2)$, Fig. 2A. The lone-pair electrons are $0.54 \AA$ from the nucleus. $\mathrm{Bi}(4)$ is coordinated to $\mathrm{O}(1), \mathrm{O}(3), \mathrm{O}(4)$, and $\mathrm{O}(9)$ whose sites are fully occupied and to one $\mathrm{O}(11)$ sof 0.76 , or $\mathrm{O}(17)$ sof 0.12 , and to one of the three ions $\mathrm{O}(13)$ sof $0.54, \mathrm{O}(18)$ sof 0.29 , or $\mathrm{O}(19)$ sof 0.16 , Fig. 2 A . The sof's in each case were restrained to sum to unity. The lone pair is $1.1 \AA$ from the nucleus indicating the largest extent of $s-p$

TABLE 5
(A) Bond Lengths ( A ) and (B) Selected Angles (deg) for $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathbf{P}_{4} \mathbf{O}_{42}$


TABLE 5-Continued

| (B) Selected Angles |  |  |  |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{O}(3)-\mathrm{Bi}(2)-\mathrm{O}(6)$ | 78.3(4) | $\mathrm{O}(15)$ \# 19-P(2)-O(18) \# 10 | 101.9(22) | $\mathrm{O}(3) \# 1-\mathrm{Pb}(3)-\mathrm{O}(3) \# 23$ | 116.6(4) |
| $\mathrm{O}(8)-\mathrm{Bi}(2)-\mathrm{O}(8) \# 1$ | 78.6(6) | $\mathrm{O}(18) \# 10-\mathrm{P}(2)-\mathrm{O}(18) \# 9$ | 116.7(35) | $\mathrm{O}(3) \# 1-\mathrm{Pb}(3)-\mathrm{O}(8) \# 23$ | 175.0(4) |
| $\mathrm{O}(3)-\mathrm{Bi}(2)-\mathrm{O}(8) \# 1$ | 86.8(4) | $\mathrm{O}(15) \# 19-\mathrm{P}(2)-\mathrm{O}(12)$ \# 18 | 102.9(22) | $\mathrm{O}(2)-\mathrm{Pb}(3)-\mathrm{O}(8) \# 1$ | 102.2(4) |
| $\mathrm{O}(6)-\mathrm{Bi}(2)-\mathrm{O}(8) \# 1$ | 164.6(5) | $\mathrm{O}(18) \# 10-\mathrm{P}(2)-\mathrm{O}(12)$ \# 18 | 115.1(19) | $\mathrm{O}(5)$ \# 1-Pb(3)-O(8) \# 1 | 64.3(3) |
|  |  |  |  | $\mathrm{O}(3) \# 1-\mathrm{Pb}(3)-\mathrm{O}(8) \# 1$ | 63.6(3) |
| $\mathrm{O}(4)$ \# 4-Bi(3)-O(7) | 80.2(4) | $\mathrm{O}(15) \#$ 19-P(2)-O(19) \# 10 | 103.1(31) | $\mathrm{O}(8)$ \# 23-Pb(3)-O(8) \# 1 | 115.7(6) |
| $\mathrm{O}(4)$ \# 4-Bi$(3)-\mathrm{O}(6) \# 1$ | 78.4(4) | $\mathrm{O}(19) \# 10-\mathrm{P}(2)-\mathrm{O}(18) \# 9$ | 110.0(54) | $\mathrm{O}(2)-\mathrm{Pb}(3)-\mathrm{O}(9) \# 6$ | 128.8(6) |
| $\mathrm{O}(7)-\mathrm{Bi}(3)-\mathrm{O}(6) \# 1$ | 102.9(4) | $\mathrm{O}(15) \# 19-\mathrm{P}(2)-\mathrm{O}(20)$ \# 19 | 103.5(22) | $\mathrm{O}(5) \# 1-\mathrm{Pb}(3)-\mathrm{O}(9) \# 6$ | 153.1(5) |
| $\mathrm{O}(4) \# 4-\mathrm{Bi}(3)-\mathrm{O}(7) \# 4$ | 90.2(3) | $\mathrm{O}(19) \# 10-\mathrm{P}(2)-\mathrm{O}(20)$ \# 19 | 117.3(29) | $\mathrm{O}(3)$ \# 1-Pb(3)-O(9) \# 6 | 71.5(3) |
| $\mathrm{O}(7)-\mathrm{Bi}(3)-\mathrm{O}(7) \neq 4$ | 79.7(4) |  |  | $\mathrm{O}(8)$ \# 1-Pb(3)-O(9) \# 6 | 104.3(4) |
| $\mathrm{O}(6)$ \# 1-Bi$(3)-\mathrm{O}(7) \neq 4$ | 167.6(4) | $\mathrm{O}(14)$ \# 18-P(2)-O(13) \# 29 | 109.0(7) | $\mathrm{O}(2)-\mathrm{Pb}(3)-\mathrm{O}(1) \# 6$ | 77.3(4) |
|  |  | $\mathrm{O}(13) \# 29-\mathrm{P}(2)-\mathrm{O}(13)$ \# 30 | 112.3(12) | $\mathrm{O}(5) \# 1-\mathrm{Pb}(3)-\mathrm{O}(1) \# 6$ | 155.4(4) |
|  |  | $\mathrm{O}(14)$ \# 18-P(2)-O(12) \# 18 | 104.2(8) | $\mathrm{O}(3) \# 1-\mathrm{Pb}(3)-\mathrm{O}(1) \# 6$ | 58.3(2) |
|  |  | $\mathrm{O}(13) \# 29-\mathrm{P}(2)-\mathrm{O}(12)$ \# 18 | 111.0(8) | $\mathrm{O}(8) \# 1-\mathrm{Pb}(3)-\mathrm{O}(1) \# 6$ | 121.5(3) |
|  |  |  |  | $\mathrm{O}(9) \# 6-\mathrm{Pb}(3)-\mathrm{O}(1) \# 6$ | 51.5(5) |

[^1]hybridization. $\mathrm{Bi}(5)$ is coordinated to the fully occupied sites by $\mathrm{O}(5), \mathrm{O}(7), \mathrm{O}(8)$; to one of the disordered $\mathrm{O}(11)$ sof 0.76 , or $\mathrm{O}(16)$ sof 0.24 ; to one of $\mathrm{O}(12)$ sof 0.42 , or $\mathrm{O}(19)$ sof 0.16 ; and to one $\mathrm{O}(13)$ sof 0.54 , or $\mathrm{O}(15)$ sof 0.23 . The lone pair is $0.27 \AA$ from the nucleus, Fig. 2B. The coordination polyhedra for $\mathrm{Bi}(2)$ to $\mathrm{Bi}(5)$ consist essentially of severely distorted octahedra regardless of which set of disordered oxygen atoms is considered as bonding. The valence bond sums $3.13,3.10,3.31$, and 3.39 strongly point to sole site occupancy by Bi.

The coordination polyhedron around $\mathrm{Pb}(1)$ is very symmetric. The cation is in the plane formed by four $\mathrm{O}(6)$ atoms that are $2.510(11) \AA$ from $\mathrm{Pb}(1)$. The polyhedron can be

TABLE 6
Lone-Pair Positions and Distances from the Nuclei, $\AA$

| Atom | $x$ | $y$ | $z$ | $d$ |
| :--- | :--- | :--- | :--- | :--- |
| $\operatorname{Bi}(1)$ | 0.5 | 0.1582 | 0.5 | 0.21 |
| $\operatorname{Bi}(2)$ | 0.1694 | 0.1329 | 0.1748 | 0.41 |
| $\operatorname{Bi}(3)$ | 0.3234 | 0.3740 | 0.3270 | 0.54 |
| $\operatorname{Bi}(4)$ | 0.6096 | 0.3041 | 0.9688 | 1.1 |
| $\operatorname{Bi}(5)$ | -0.0197 | 0.3224 | 0.34 | 0.27 |
| $\operatorname{Pb}(1)$ | 0 | 0 | 0 | - |
| $\operatorname{Pb}(2)$ | -0.2892 | 0 | 0.3494 | 0.56 |
| $\operatorname{Pb}(3)$ | 0.2972 | 0 | 0.3070 | 0.68 |

completed by four $\mathrm{O}(18)$ atoms at $3.10(5) \AA$, sof 0.29 , forming an elongated cube, Figs. 2B and 3, or by two $\mathrm{O}(14)$, sof 0.27 , at $2.185(1)$ or by two $\mathrm{O}(20)$, sof 0.08 , at $2.394(1)$, each set forming an octahedron, Fig. 2B. The calculated lone-pair location, Table 6 , indicates that the $6 s^{2}$ electrons are spherical. $\mathrm{Pb}(2)$ is coordinated to seven fully occupied oxygen sites and $24 \%$ of the time $\mathrm{O}(17)$ is present, Fig. 2B. The lone pair is $0.56 \AA$ from the nucleus. The polyhedron formed by the eight fully occupied anion sites around $\mathrm{Pb}(3)$ can be considered as a distorted monocapped pentagonal bipyramid, Fig. 2B. It is the same as for $\mathrm{Pb}(2)$ when $\mathrm{O}(17)$ is present. The lone pair is $0.68 \AA$ from the nucleus of $\mathrm{Pb}(3)$. The valence bond sums $2.50,2.54$, and 2.64 may indicate that $\operatorname{Bi}(1)$ partially occupies these three sites.
$\mathrm{P}(1)$ and $\mathrm{P}(2)$ atoms occupy the centers of tetrahedra. Two fairly regular tetrahedra exist around $\mathrm{P}(1)$ : $\mathrm{P}(1)-\mathrm{O}(9)-$ $\mathrm{O}(10)-2 \mathrm{O}(11)$ and $\mathrm{P}(1)-\mathrm{O}(9)-\mathrm{O}(17)-2 \mathrm{O}(16)$. Based on the sof of the oxygen anions the first tetrahedron is present about $76 \%$ of the time and the second $24 \%$. The two tetrahedra have as a common apex $\mathrm{O}(9)$. The tetrahedra are rotated about the bond $\mathrm{P}(1)-\mathrm{O}(9)$ by $55^{\circ}$, Fig. 1. Three somewhat less regular tetrahedra are present around $\mathrm{P}(2)$. The first is formed by $\mathrm{O}(12)-\mathrm{O}(15)-2 \mathrm{O}(18)$, the second by $\mathrm{O}(15)-\mathrm{O}(20)-2 \mathrm{O}(19)$, and the third by $\mathrm{O}(12)-\mathrm{O}(14)-2 \mathrm{O}(13)$, Fig. 1. The first and second tetrahedra are related by a $55^{\circ}$ rotation about $\mathrm{P}(2)-\mathrm{O}(15)$. The third tetrahedron is obtained by a rotation of about $55^{\circ}$ around $\mathrm{P}(2)-\mathrm{O}(12)$ starting


FIG. 2. Oxygen coordination polyhedra around (A) $\operatorname{Bi}(1), \operatorname{Bi}(2), \operatorname{Bi}(3)$, and $\operatorname{Bi}(4)$ and $(\mathrm{B}) \mathrm{Bi}(5), \mathrm{Pb}(1), \mathrm{Pb}(2)$, and $\mathrm{Pb}(3)$. When equivalent bond lengths occur they are omitted. When more than one set of disordered oxygen atoms are possible only one is illustrated for the sake of clarity. However, all possible bonding oxygen atoms to $\mathrm{Pb}(1)$ are shown. LP represents the displacement of the lone-pair electron density from the cation nucleus.
from the first tetrahedron. All of these oxygen atoms exhibit partial occupancy as shown in Table 3. Of course, only one of each tetrahedron centered on $\mathrm{P}(1)$ and $\mathrm{P}(2)$ is present in any one unit cell and the disorder is most likely statistical creating domains within the crystal. It is not possible to fix uniquely which of the tetrahedra are simultaneously present in a given unit cell. However, based on the oxygen site occupancies the tetrahedron $\mathrm{P}(1)-\mathrm{O}(9)-\mathrm{O}(10)-2 \mathrm{O}(11)$ occurs much more frequently than $\mathrm{P}(1)-\mathrm{O}(9)-\mathrm{O}(18)-2 \mathrm{O}(17)$. Similarly the most frequently occurring tetrahedron for $\mathrm{P}(2)$ is $\mathrm{P}(2)-\mathrm{O}(12)-\mathrm{O}(14)-2 \mathrm{O}(13)$ and it is reasonable to assume that the most frequently occurring tetrahedra are present at the same time in a given unit cell. Selected bond lengths and angles are shown in Table 5. The interatomic distances among $\mathrm{O}(1)$ to $\mathrm{O}(9)$ are greater than $2.6 \AA$; bond distances among oxygen atoms with partial occupancies when they are simultaneously present range from 2.3 to $2.6 \AA$; unrealistically short distances, e.g., $1.5 \AA$, are calculated for disordered oxygen atoms that cannot simultaneously be present.

Units of $\mathrm{Pb}_{2} \mathrm{O}_{2}$ parallel to [100] result from the sharing of the polyhedral edge formed by $\mathrm{O}(2)$ and $\mathrm{O}(5)$ that are bonded to both $\mathrm{Pb}(2)$ and $\mathrm{Pb}(3)$. Two such $\mathrm{Pb}_{2} \mathrm{O}_{2}$ units bond parallel to [001] by forming a weak van der Waals bond $\mathrm{Pb}(3)-\mathrm{O}(1), \quad 3.147(10) \AA$, and a strong bond $\mathrm{Pb}(2)-\mathrm{O}(1), 2.436(10) \AA$, to create $\mathrm{Pb}_{4} \mathrm{O}_{6}$ entities, Fig. 4. These tetrameric entities are separated by $\mathrm{P}(1) \mathrm{O}_{4}$ tetrahedra
and by two separate $\mathrm{P}(2) \mathrm{O}_{4}-\mathrm{Pb}(1)-\mathrm{P}(2) \mathrm{O}_{4}$ columns parallel to [100] that are repeated by the $c$ axis periodicity, Fig. 4. This layer parallel to the $a-c$ plane is sandwiched between


FIG. 3. $\mathrm{Pb}(1)-\mathrm{Bi}-\mathrm{O}$ articulations parallel to $(010)$. The $\mathrm{Pb}(1)-\mathrm{O}$ coordination polyhedron is an elongated cube. Only one possible orientation of $\mathrm{P}(2) \mathrm{O}_{4}$ is depicted.


FIG. 4. Crystal structure of $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathrm{P}_{4} \mathrm{O}_{42}$ looking along [010]. The heavy line shows the $\mathrm{Pb}(2)-\mathrm{O}(1) 2.436(10) \AA$ bond and the dotted line the long $\mathrm{Pb}(3)-\mathrm{O}(1) 3.147(10) \AA$ bond for the $\mathrm{Pb}_{4} \mathrm{O}_{6}$ moiety.
the $\mathrm{Bi}-\mathrm{O}$ layers that are parallel to the $a-c$ plane and perpendicular to [010], Fig. 3. They are interconnected along [100] by $\mathrm{P}(2) \mathrm{O}_{4}-\mathrm{Pb}(1)-\mathrm{P}(2) \mathrm{O}_{4}$, Fig. 3. The two-dimensional sheets perpendicular to [010] are fused by inter-


FIG. 5. Crystal structure of $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathrm{P}_{4} \mathrm{O}_{42}$ looking along [001].
layer $\mathrm{P}(1) \mathrm{O}_{4}$ groups, $\mathrm{Pb}(2)$, and $\mathrm{Pb}(3)$ into a three-dimensional network, Fig. 5.

The unit cell parameters of $\mathrm{Pb}_{5} \mathrm{Bi}_{18} \mathrm{P}_{4} \mathrm{O}_{42}$ are related to those of the fluorite type $\delta-\mathrm{Bi}_{2} \mathrm{O}_{3}$ by the transformation $a=\frac{3}{2} a_{\mathrm{f}}-\frac{3}{2} b_{\mathrm{f}} ; b=\frac{3}{2} a_{\mathrm{f}}+\frac{3}{2} b_{\mathrm{f}} ; c=3 c_{\mathrm{f}}$. This corresponds approximately to a $3 \times 3 \times 3$ superstructure of a smaller tetragonal I-centered cell, $a=b=3.9 \AA$ an $c=5.5 \AA$, which is based on the cubic F-centered fluorite cell $a=5.525 \AA$. The atoms $\mathrm{Bi}, \mathrm{Pb}$, and P occupy nearly the same positions as Bi in $\delta-\mathrm{Bi}_{2} \mathrm{O}_{3}$. Numerous compounds adopt the same type of superstructure (14), Kashida et al. (11) found partial disorder of oxygen ions coordinated to V in $\mathrm{Bi}_{2} \mathrm{O}_{3}: \mathrm{V}_{2} \mathrm{O}_{5}=9: 1$.

## ACKNOWLEDGMENTS

S.G. and H.S. gratefully acknowledge the support of the R. A. Welch Foundation of Houston, TX. S.G. thanks the IMCCEC/US consortium under Contract No. W-31 for a travel grant. Portions of this work were performed at the DuPont-Northwestern-Dow Collaborative Access Team (DND-CAT) Synchrotron Research Center located at Sector 5 of the Advanced Photon Source. DND-CAT is supported by the E.I. DuPont de Nemours \& Co., The Dow Chemical Company, the U.S. National Science Foundation through Grant DMR-9304725 and the State of Illinois through the Department of Commerce and the Board of Higher Education Grant IBHE HECA NWU 96. Use of the Advanced Photon Source was supported by the U.S. Department of Energy, Basic Energy Sciences, Office of Energy Research, under Contract No. W-31-102-Eng-38. We thank W. Marshall from DuPont and D. Keane and Z. Wawrzak from DND-CAT for their technical assistance.

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[^0]:    Note. The anisotropic displacement exponent takes the form $-2 \pi^{2}\left[h^{2} a^{x 2} U_{11}\right.$ $\left.+\cdots+2 h k a^{x} b^{x} U_{12}\right]$.

[^1]:    Note. Symmetry transformations used to generate equivalent atoms: $\# 1-x+\frac{1}{2},-y+\frac{1}{2},-z+\frac{1}{2} ; \# 2 x+\frac{1}{2},-y+\frac{1}{2}, z+\frac{1}{2} ; \# 3,-x+1,-y$, $-z+1 ; \# 4 x+\frac{3}{2},-y+\frac{1}{2},-z+\frac{1}{2} ; \# 5 x-\frac{1}{2},-y+\frac{1}{2}, z+\frac{1}{2} ; \# 6 x-\frac{1}{2}, y-\frac{1}{2}, z+\frac{1}{2} ; \# 7-x+1, y,-z+1 ; \# 8-x+1,-y+1,-z+1 ; \# 9$ $x-\frac{1}{2}, y-\frac{1}{2}, z-\frac{1}{2} ; \# 10 x-\frac{1}{2},-y+\frac{1}{2}, z-\frac{1}{2} ; \# 11-x+1, y,-z ; \# 12 x,-y+1, z-1 ; \# 13 x+\frac{1}{2},-y+\frac{1}{2}, z-\frac{1}{2} ; \# 14 x,-y+1, z ; \# 15-x$, $y,-z+1 ; \# 16 x-\frac{1}{2}, y+\frac{1}{2}, z-\frac{1}{2} ; \# 17-x+1,-y,-z ; \# 18 x-1, y, z ; \# 19 x-1, y, z-1 ; \# 20-x,-y,-z ; \# 21+x, y,-z ; \# 22 x,-y, z$; $\# 23-x+\frac{1}{2}, y-\frac{1}{2},-z \frac{1}{2} ; \# 24-x+\frac{3}{2}, y-\frac{1}{2},-z+\frac{1}{2} ; \# 25 x+\frac{1}{2}, y-\frac{1}{2}, z-\frac{1}{2} ; \# 26 x, y, z-1 ; \# 27-x+\frac{3}{2}, y+\frac{1}{2},-z+\frac{1}{2} ; \# 28-x+1,-y+1$, $-z$; \#29 $x-1, y-1, z-1 ; \# 30 x-1,-y+1, z-1 ; \# 31 x+\frac{1}{2}, y+\frac{1}{2}, z-\frac{1}{2} ; \# 32 x+1, y, z ; \# 33-x+\frac{1}{2}, y+\frac{1}{2},-z+\frac{1}{2} ; \# 34-x+\frac{3}{2}, y+\frac{1}{2}$, $-z+\frac{3}{2} ; \# 35-x+\frac{3}{2},-y+\frac{1}{2},-z+\frac{3}{2} ; \# 36 x, y, z+1 ; \# 37 x+\frac{1}{2}, y+\frac{1}{2}, z+\frac{1}{2} ; \# 38 x, y-1, z-1 ; \# 39 x, y+1, z ; \# 40 x+1, y+1, z+1 ; \# 44 x$, $y+1, z+1 ; \# 42 x,-y+2, z ; \# 43 x+\frac{1}{2},-y+\frac{3}{2}, z+\frac{1}{2} ; \# 44 x,-y+1, z 1 ; \# 45 x+1, y, z+1 ; \# 46 x, y-1, z ; \# 47 x+\frac{1}{2}, y-\frac{1}{2}, z+\frac{1}{2} ; \# 48$ $-x+\frac{3}{2}, y-\frac{1}{2},-z+\frac{3}{2} ; \# 49 x-\frac{1}{2}, y+\frac{1}{2}, z+\frac{1}{2} ; \# 50 x-\frac{1}{2},-y+\frac{3}{2}, z-\frac{1}{2}$. ${ }^{a}$ The valence bond values for disordered atoms were calculated by multiplying v.b. by sof.

